

Starch/PVA-based nanocomposites reinforced with bamboo nanofibrils

Mario Guimarães Jr.^{a,*}, Vagner Roberto Botaro^b, Kátia Monteiro Novack^c,
Fábio Gomes Teixeira^d, Gustavo Henrique Denzin Tonoli^e

^a Departamento de Eletromecânica – Centro Federal de Educação Tecnológica de Minas Gerais, Araxá, Doctoral Candidate in Materials Engineering for REDEMAT/UFOP, Av. Ministro Olavo Drummond, 25, São Geraldo, Araxá, MG CEP 38180-510, Brazil

^b Departamento de Pós-Graduação em Ciência dos Materiais – Universidade Federal de São Carlos, Sorocaba, Cx.P 3031, Sorocaba, SP CEP 18052-780, Brazil

^c Instituto de Ciências Exatas e Biológicas/Departamento de Química–Universidade Federal de Ouro Preto, Ouro Preto, MG, Campus Universitário, morro do Cruzeiro, CEP 35400-000, Brazil

^d Instituto de Tecnologia de Alimentos, Centro de Tecnologia de Embalagem (ITAL/CETEA), Avenida Brasil, 2880, Campinas, SP CEP 13070-178, Brazil

^e Departamento de Ciências Florestais, Universidade Federal de Lavras, Cx P. 3037, Lavras, MG CEP 37200-000, Brazil

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ABSTRACT

This work aimed to evaluate the effect of including different concentrations of bamboo nanofibrils on physical, mechanical, morphological and structural properties of nanocomposites from cassava starch and polyvinyl alcohol (PVA). Nanocomposites were prepared with blends of starch/PVA and nanofibrils of bamboo. Chemical pre-treatments and mechanical defibrillation were used to obtain the nanofibrils. The mixture containing 3% of starch and 4% of PVA in the proportion of 20/80 (starch/PVA) were chosen after preliminary testing. Atomic force microscopy (AFM) and transmission electronic microscopy (TEM) were used to characterize the bamboo nanofibrils. Microstructure of the nanocomposites was evaluated using scanning electron microscopy (SEM) and X-ray diffractometry (XRD). Physical and mechanical properties were also evaluated. Results showed that pre-chemical treatments increased the content of the alpha-cellulose in bleached pulp by approximately 112% in relation to the native fiber. Increasing the number of passages through the defibrillator reduced the average diameter of the bamboo nanofibrils (from 82 ± 29 nm to 10 ± 6 nm). Addition of 6.5% nanofibrils improved the tensile strength and elongation at the break of the nanocomposite by 24 and 51%, respectively, but reduced the tensile modulus by 40% in relation to control (unreinforced) blend. Nanofibrils decreased the transparency of the nanocomposite films. The water vapor permeability and water solubility of the nanocomposite containing high contents of nanofibrils decreased up to 20% and 30%, respectively, in relation to the control blend.

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1. Introduction

Ecological concern has increased the interest on natural and renewable materials, making plant fibers and biodegradable polymers an interesting and secure alternative for development of new materials. Most common biodegradable polymers are obtained from polysaccharides and polyesters. Between the polysaccharides, thermoplastic starch (TPS) has been largely used, despite presenting some disadvantages, such as strong hydrophilicity, poor mechanical properties in comparison to conventional polymers and variations of post-processing properties. Some strategies have been presented to improve these properties, including the use

of blending (Mohanty et al., 2000; Averous and Boquillon, 2004; Olobarrieta, 2005).

Several studies have examined blending starch with other polymers due to the relative facility of obtaining materials with desirable properties with no need to significant changes or investments on the conventional process. Thus, polymeric blends are a versatile technological solution to obtain polymeric materials with myriad specifications at relatively low cost using combinations of polymers with properties of interest (Utracki, 1989; Ishiaku et al., 2002). Many types of biodegradable PVA-based composites have been prepared through mixture with starch (Follain et al., 2005; Lawton, 1996; Siddaramaiah et al., 2004; Cinelli et al., 2006; Zhai et al., 2003; Khan et al., 2006). Chemical modification, including cross linking with glutaraldehyde and epichlorohydrin, has also been used by several authors to improve the physical properties

* Corresponding author. Tel.: +55 34 3669 4500.

E-mail address: mgjunior@araxa.cefetmg.br (M. Guimarães Jr.).

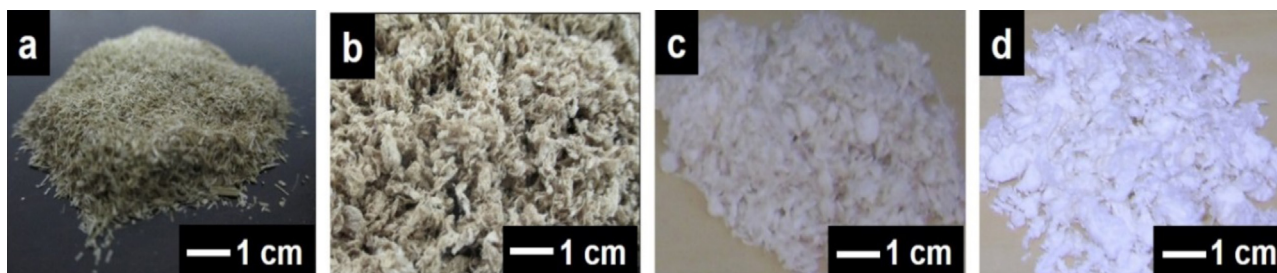


Fig. 1. Visual appearance: (a) ground bamboo native fiber; (b) refined pulp without bleaching; (c) refined pulp bleached for one time (1×); and (d) refined pulp bleached for two times (2×).

and limitations of these polymers (Nabar et al., 2006; Beliakova et al., 2004; Ramaraj, 2007; Sreedhar et al., 2006).

In recent years, nanofibrils and nanocrystals obtained from plant fibers have been studied as reinforcement in starch-based nanocomposites. The use of these two types of cellulosic nanoreinforcements in polysaccharides matrices has increased because of their great affinity. Large specific surface area, high aspect ratio and high capacity to form structured nets make cellulosic nanofibrils excellent low-cost and non-toxic reinforcement agents (Chang et al., 2010; Chang et al., 2010). This class of materials presents micrometric lengths and nanometric widths, leading to interest potential for several applications, including drugs (Villanova et al., 2011), dietary food (Okiyama et al., 1993), special papers (Nguyen and Tan, 2009), reinforcement in packaging/films (Siro and Plackett, 2010; Syverud et al., 2011) and polymeric matrices (Orts et al., 2005; Azizi Samir et al., 2005; Siqueira et al., 2010).

The present work aimed to evaluate the effect of including various concentrations of bamboo nanofibrils on mechanical, morphological, structural and physical properties of cassava starch/PVA-based nanocomposites.

2. Materials and methods

2.1. Materials

Commercial unbleached refined bamboo cellulose pulps from *Bambusa Vulgaris* Schrad (around 2 years of age), were obtained from CEPASA – Celulose e Papel de Pernambuco S/A, Jaboatão dos Guararapes, Brazil. The bamboo pulp was produced by the soda-anthraquinone process (NaOH-AQ), with approximately, 18% NaOH and 0.03% anthraquinone ($C_{14}H_8O_2$) per unit of solid mass, pH between 12 and 13, and with a fiber yield estimated in 46%. After batching in a Pandia continuous digester at 6–7 bar and 170 °C for 45 min, samples were disk refined until a Schopper Riegler (SR) index of around 25–30.

Composite matrix was formed by: modified cassava starch – FMM (Cargill, lot 1,209,035, type A), crystallinity index (CI) of around 45% and amylopectin content of 85%; polyvinyl alcohol – PVA (Sigma–Aldrich, lot MKBK3473V), with molecular weight (Mw) of 130,000 g/mol, 99% hydrolyzate; and liquid glycerol (Sigma–Aldrich) as plasticizer agent, with Mw of 92.09 g/mol, ≥ 99% concentration and density of 1.26 g/mL.

2.2. Pre-treatment and characterization of the bamboo pulp

Before obtaining bamboo nanofibrils, the refined pulp sample was subjected twice to alkaline treatment as proposed elsewhere (Corradini et al., 2006) followed by bleaching (Pereira et al., 2010) but maintaining the pH at 11 in order to prevent brightness reversion. Before repeating or starting a new treatment, samples were washed with tap water until the pH becomes neutral. In order to calculate fiber yield, the samples were oven dried at 60 °C for 24 h and weighted before and after each treatment.

Chemical composition of the refined pulps, with and without treatments, was determined. Sample preparation for chemical analysis followed the procedures described in T264 cm-97 (TAPPI, 1999a) and T257 cm-97 (TAPPI, 1999b) standards. The percentage of holocellulose (cellulose + hemicelluloses) was determined as described in T9 m-54 (TAPPI, 1999c). The contents of alpha-cellulose and lignin (insoluble in acid) were estimated according to T203 cm-99 (TAPPI, 1999d) and T222 cm-88 (TAPPI, 1999e) standards, respectively. The content of hemicelluloses was calculated from the difference between the values of holocellulose and alpha-cellulose, and the contents of ash and extractives were calculated in accordance with T211 cm-93 (TAPPI, 1999f) and T204 cm-97 (TAPPI, 1999g) standards, respectively. The procedure for determining solubility in ethanol–toluene (1:2, v/v) was conducted for 6 h in ethanol/toluene and 4 h in ethanol. All determinations were performed in triplicate.

After pre-treatment and characterization, the native fiber and refined pulp without bleaching were called FNA and PRST, respectively, and the pulp twice alkaline treated and bleached were called PRM2 and PRM2B2, respectively. Fig. 1a–d show the ground bamboo native fiber, refined pulp without bleaching, refined pulp bleached for one time (1×) and refined pulp bleached for two times (2×).

2.3. Production of bamboo nanofibrils

Bamboo nanofibrils were obtained by mechanical defibrillation of the pre-treated (twice alkaline treated and twice bleached) refined pulp fibers using a Super Masscolloider Masuko Sangyo MKCA6-3 defibrillator at 1500 rpm (Ifuku et al., 2010). Initially, the bamboo pulp was immersed (at 1.2% w/v concentration) in water for 48 h for hydration to guarantee the swelling of the fiber cell wall. The gap between the silicon carbide stones in the defibrillator was adjusted to 100 μm (Nakagaito and Yano, 2004) and suspensions from refined/bleached fibers were fed several times until form a gel, which began to happen after 15 passages through the defibrillator (Bufalino et al., 2014; Guimarães Junior et al., 2015). The number of passages adopted for this study was 5 and 30, and the electric current consumed was kept at 6 A. Sonication was applied after defibrillation, for 30 min, using a Branson sonicator with a 13 mm tip, at 450 W (25% amplitude) and 20–25 KHz, which led to 840 J/mL energy.

The suspensions were called NRB5x and NRB30x for nanofibrils obtained after 5 and 30 passages through the defibrillator, respectively. Fig. 2 shows the steps of production the bamboo nanofibrils from the solution of bleached refined pulp until obtaining the gel after 30 passages through the mechanical defibrillator.

2.4. Characterization of the bamboo nanofibrils

Transmission electron microscopy (TEM) was performed in order to evaluate the morphology of bamboo nanofibrils, using a

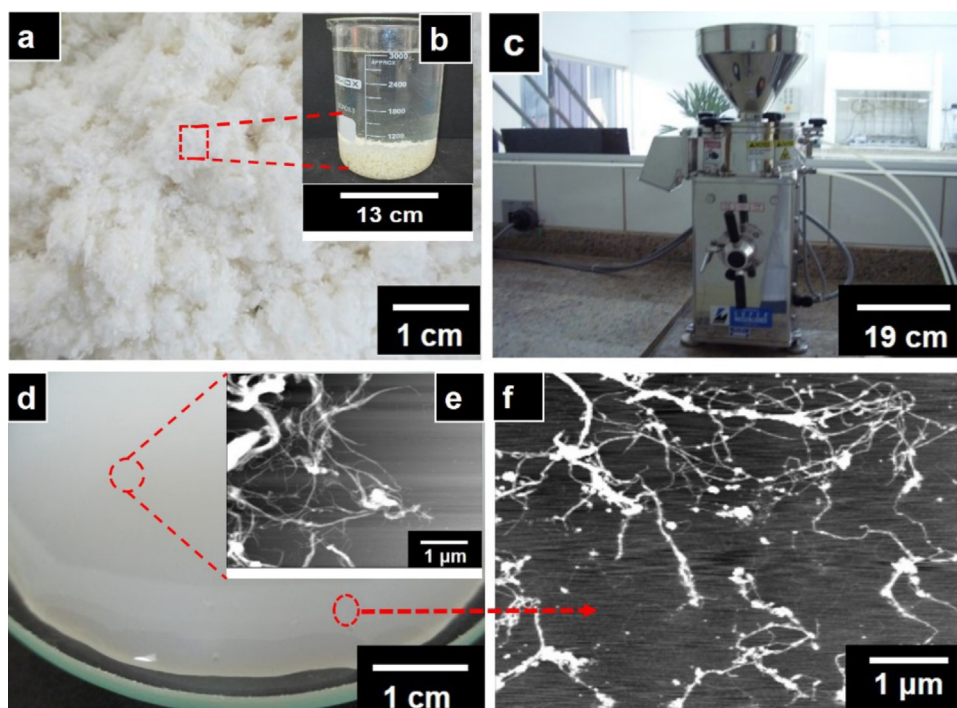


Fig. 2. Stages of production of bamboo nanofibrils: (a) refined and twice bleached bamboo pulp (Guimarães Junior et al., 2015); (b) aqueous solution of the same bamboo pulp; (c) Super Masscolloider Masuko Sangyo MKCA6-3 defibrillator; (d) visual appearance of the gel formed after 30 passages of the solution of bamboo pulp through the defibrillator (Guimarães Junior et al., 2015); (e) and (f) AFM micrographs of bamboo nanofibrils.

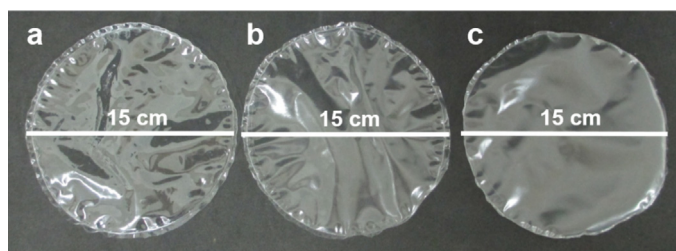


Fig. 3. (a) FMM/PVA blend film (P80/A20); (b) nanocomposite with bamboo nanofibrils (6.5% dry mass) obtained after 5 passages through the defibrillator; (c) nanocomposite with nanofibrils (6.5% dry mass) obtained after 30 passages through the defibrillator. Diameter of the films = 15 cm.

Carl Zeiss EM 912 instrument with corrected Omega filter peels, at tension of about 100 kV. Aqueous solutions of cellulose nanofibrils (0.05 g/L) stained with uranyl acetate were deposited and dried on copper microgrids of 400 mesh. Nanofibril dimensions were measured using the Image J 1.47 V software. At least 100 measurements were randomly performed to determine the average diameter of the nanofibrils.

Atomic force microscopy (AFM) was performed with an Agilent 5500 N9410S microscope. A drop of a dilute suspension of approximately 0.05 g/L was deposited on a cleaved mica surface and dried at 60 °C for 12 h. The images were obtained in dynamic mode at room temperature with a scan rate of 1 Hz, using silicon tips with a curvature radius of less than 10 nm and spring constant of 42 N/m. Image processing and diameter measurements were performed using 64-bit Gwyddion software. The diameters of the nanofibrils were estimated by measuring their heights to eliminate the convolution effect between the probe tip and the bamboo nanofibrils (Kvien et al., 2005). Around 100 nanofibrils were randomly chosen for each condition, and two height measurements were used to determine the average diameter (Silvério et al., 2013). The surface roughness of the mica surface after the deposition of the nanofib-

rils was measured in 5 μm × 5 μm scanned areas, using the average surface roughness parameter (R_a in Eq. (1)) of the 64-bit Gwyddion software (Kaboarani and Riedl, 2012). Roughness (R_a) was evaluated in order to estimate how is dispersion of nanofibrils for each number of passages through the defibrillator.

$$R_a = \frac{1}{n} \sum_{i=1}^n Z_i - Z_m, \quad (1)$$

where:

$$Z_m = \frac{1}{n} \sum_{i=1}^n Z_i \quad (2)$$

R_a (average roughness) is the arithmetic mean of the absolute deviation values from the surface height. Z_m is the average height of the sample points, and Z_i is the height of each sample point. The number of points within each area is represented by n .

2.5. Preparation of the control blend and nanocomposites

A solution of starch (FMM) was prepared with 3% starch (FMM), 12% glycerol and 85% water. A PVA solution was prepared with 4% PVA, 25% glycerol and 71% water. Both combinations were defined based on previous investigations regarding different proportions of raw materials, and on the ensuing results of water vapor permeability, mechanical and physical properties. For preparing the solutions, both FMM and PVA powders remained immersed for 24 h in distilled water under mechanical stirring at 100 rpm at room temperature for hydration. Then, glycerol was added and both solutions of FMM and PVA were submitted to magnetic stirring at 500 rpm at 80 and 90 °C, respectively for 20 min. FMM and PVA solutions were mixed in the proportion of 20/80 (FMM/PVA) at room temperature in a homogenizer at 15,000 rpm for 5 min. Around 40 g of the final solution was poured into Plexiglas acrylic plates (15 cm in diameter) for obtaining of the control sample (called P80/A20) by film casting. Finally, it was sonicated and placed in a controlled room at 20 °C and 60% relative humidity for 10 days for drying

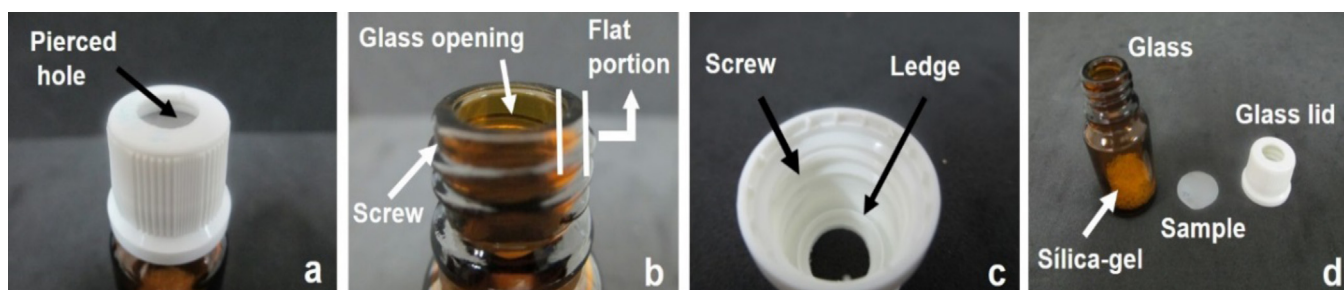


Fig. 4. Permeability cell for determination of water vapor permeability. (a) hole in the cap for diffusion of water through the specimen; (b) flat portion of contact between the cap and the vial, to avoid interferences between the pressures created; (c) plastic ledge that press the flat specimen in the flat portion of the vial; and (d) parts that permeability cell.

and moisture stabilization. The thickness of the control blend films varied between 0.064 and 0.093 mm.

The nanocomposites were also prepared by film casting with the blends of starch (FMM) and PVA at 20/80 (FMM/PVA) ratio and reinforced with four different contents (0.5, 1.5, 4.5 and 6.5% in relation to total solid mass of the film) of bamboo nanofibrils obtained with 5 and 30 passages through the defibrillator. The aqueous suspensions of nanofibrils were added to the FMM/PVA blend under mechanical stirring at 750 rpm at room temperature for 15 min, and homogenized at 15,000 rpm at room temperature for 20 min. Then, the blend solutions with nanofibrils were subjected to 1 h of sonication at 450 W with 25% amplitude (energy of around 1.7 kJ/mL). Finally, the solutions were poured into Plexiglas acrylic plates and placed in a controlled room at 20 °C and 60% relative humidity for 10 days for drying and moisture stabilization. The thickness of the nanocomposite films varied between 0.073 and 0.121 mm. Fig. 3 shows the blend films and flexible nanocomposites obtained.

2.6. Characterization of the nanocomposites

2.6.1. Mechanical properties

The maximum tensile strength (TS), the elongation at breaking (TB) and the tensile modulus (TM) of nanocomposites were determined in an INSTRON model 5966–E2 instrument with an 1 kN load cell, deflection rate of 50 mm/min, and initial distance between the clamping jaws of 50 mm. The tensile test was performed following the ASTM D 882–00 standard. Test specimens with 15 mm width and 100 mm length were used, and the TM was calculated from the tangent of the initial linear function of the stress–strain curve. Five repetitions were used for each percentage of bamboo nanofibrils tested.

2.6.2. Scanning electron microscopy (SEM)

A Shimadzu SSX-550 scanning electron microscope (SEM) operating in a high vacuum chamber at acceleration voltage of 15 kV was used to analyze the surface morphology of the blend and nanocomposite films. A secondary electron detector was used to capture the images. A nanolayer of gold particles was deposited on the film surfaces for 2 min at 330 V and 20 mA in a physical vapor depositor (PVD).

2.6.3. X-ray diffractometry (XRD)

The crystallinity index of the nanocomposites were estimated from X-ray diffraction using a Shimadzu XRD-6000 diffractometer at room temperature, step size of 0.05° and an integration time of 1 s. The radiation used was CuK α (1.5406 Å), which was generated by 40 kV and an incident current of 30 mA. Flat pieces of nanocomposites were cut and fixed in the glass sample holder. Crystallinity index was calculated by deconvolution of the diffraction peaks using a Lorentzian function of the crystalline and amorphous peaks

(Chen et al., 1999). A Savitzky-Golay filter was used to smooth the curves.

2.6.4. Physical properties

2.6.4.1. Optical transmittance (Tr). The transmission of light through the nanocomposites was measured in a Shimadzu UV–vis 1601 spectrophotometer, with a double-beam tungsten lamp, accuracy of 0.5 nm and a spectral band of 2 nm. This technique allowed the evaluation of the total percentage of incident light that is transmitted through the nanocomposites before and after the incorporation of bamboo nanofibrils. The scan was performed between 400 and 800 nm, and transparency was evaluated according to procedures described elsewhere, (Kampeerappun et al., 2007; Chen et al., 2008). Specimens were cut into rectangles with 10 mm \times 50 mm and adhered to the internal wall of the quartz cuvette of the spectrophotometer (Park and Zhao, 2004). Three repetitions were performed for each nanofibril content. The empty quartz cuvette was used as the totally transparent control. The device was programmed to display the light transmittance value (%) passed through the nanocomposites.

2.6.4.2. Water vapor permeability (WVP). The water vapor transmission rate (WVTR) of the blend and nanocomposites was determined by gravimetry using the ASTM E 96–00 standard. The test specimens were cut with a radius of 5.25 mm and placed between the cap and the amber glass vials of the permeability cell (Fig. 4) with $\frac{3}{4}$ of its volume containing CaCl₂ (desiccant) previously dried. Then, the amber glass was placed in hermetic desiccators at 18.5 ± 2 °C, whose a volume of water sets the water activity to 0.1 in contact with the upper face of the specimen. The weight gain was measured each 24 h until 10 days of moisture exposure. A plot of weight gain versus time was used to determine the WVTR. The slope of the linear portion of this plot represents the steady state amount of water vapor diffusing through the specimen. Three repetitions were performed for blend and each nanofibril content. The water vapor permeability (WVP) was calculated by multiplying the steady WVTR by the specimen thickness (mm) and dividing that by the water vapor pressure difference ($\Delta p = 2.1297$ kPa; Tetens, 1930) across the specimen (Bourtoom and Chinnan, 2008). Fig. 4 shows details of the permeability cell.

2.6.4.3. Solubility. Water solubility was determined using circular test specimens of 3.14 cm². The initial dry mass was obtained after drying at 55 ± 2 °C for 24 h. Specimens were immersed in a vessel containing 30 mL of distilled water and magnetic stirred at 50 rpm for 24 h at room temperature. The resultant suspensions were filtered, and the residues were dried at 105 °C for 24 h. After weighing, the amount of non-solubilized material was determined. The solubility of the specimens was expressed as percentage of mass solubilized in relation to the initial mass (Gontard et al., 1994).

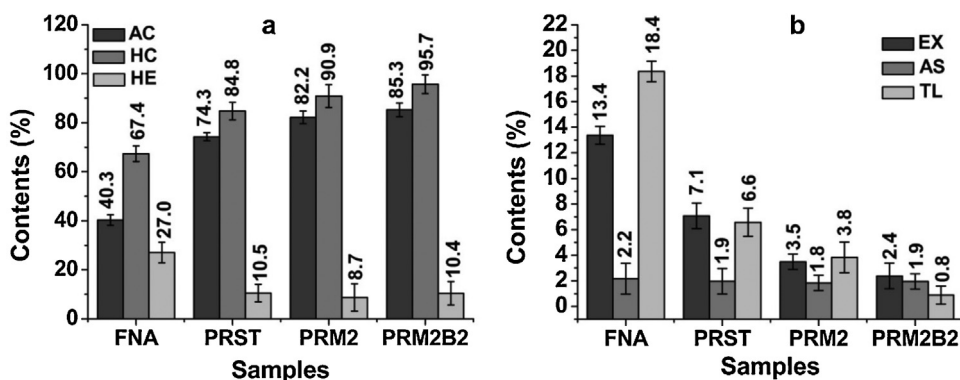


Fig. 5. Average and standard deviation values of chemical constituents of the bamboo native fiber (FNA), refined bamboo pulp without pre-treatments (PRST), refined bamboo pulp twice treated with NaOH (PRM2), refined bamboo pulp twice treated with NaOH and twice bleached (PRM2B2). AC=alpha-cellulose; HC=holocellulose; HE=hemicellulose; EX=extractives; AS=ashes; and TL=total lignin.

2.6.5. Statistical analysis

Sisvar 5.0 was used for carrying out statistical analyzes of data through average comparison. Fisher's least significant difference and Scott–Knott test were used at 95% confidence level for mechanical and physical properties.

3. Results and discussion

3.1. Characterization of the native fiber and bamboo pulp

Lignin and hemicelluloses removal from surrounding the cellulose micro/nanofibrils of the plant cell wall is one of the most important steps for obtaining cellulose nanofibrils, since lignin and hemicelluloses act as binders in the fibrillar structure, preventing their individualization (Nakagaito and Yano, 2004; Dufresne et al., 1997; Dinand et al., 1999). Purely mechanical processes can cause excessive cellulose fibril breaking and incur an increase in energy consumption (Siro and Plackett, 2010). The combination of both chemical and mechanical methods can lead to nanofibrils with excellent quality and with low energy consumption.

Fig. 5 presents the chemical constitution of the fibers after the pre-treatments. The native bamboo fiber (FNA) presented lower percentages of alpha-cellulose (AC=40.3%) and holocellulose (HC=67.4%), and higher percentages of hemicelluloses (HE=27.0%), total lignin (TL=18.4%), extractives (EX=13.4%) and ash (AS=2.2%) compared to the pre-treated pulps. After pulping with NaOH-AQ (sodium hydroxide + anthraquinone) followed by refining (PRST), the content of alpha-cellulose (AC) and holocellulose (HC) of the pulp increased significantly to 74.3% and 84.8%, respectively, while the content of hemicelluloses (HE), total lignin (TL) and extractives (EX) decreased to 10.5, 6.6 and 7.1%, respectively. Results agree with values found in literature for the same studied species (Gomide et al., 1982; Morais et al., 2000; Gomide, 1986) and within the range for other bamboo species with age between 2 and 4 years (Mwaikambo, 2006; Li et al., 2007; Wahab et al., 2013).

After the second bleaching (PRM2B2), there was an increase of 111.7 and 42.0% in the content of AC and HC, while values of HE, TL and EX reduced by 61.5, 96.0 and 82.1%, respectively in relation to natural fiber (FNA). The high content of AC and HC after pre-treatment indicate the opening of the lignocellulosic structure of the fibers, causing hydrolysis of the hemicelluloses and cleavage of lignin-hemicelluloses bonds, which resulted in removal of hemicelluloses and almost all the lignin of the pulp (Suess, 2010).

3.2. Characterization of the bamboo nanofibrils

3.2.1. Morphological analyses

Fig. 6a and b depicts AFM images of the bamboo nanofibrils obtained after 5 and 30 passages through the defibrillator, respectively. The mechanical disruption caused by defibrillation resulted in fibrillar structures with lengths estimated on tens of micrometers and diameters predominantly smaller than 100 nm (Fig. 6c and d). The reduction of the average diameter with the increase of number of passages (i.e., from 5 to 30) was of around 88%. The diameters changed from 82 ± 29 nm to 10 ± 6 nm, turning the nanofibril suspensions more stable for the sample with 30 passages through the defibrillator. Furthermore, the suspension with 30 passages presented 66% of nanofibrils with diameter between 5 and 15 nm (Fig. 6d), while 5 passages let to around 37% of nanofibrils in the same range (Fig. 6a). The analysis of roughness (Ra) of the surface after nanofibril deposition showed high values of Ra for samples obtained with 5 passages, which is evidence of low quality dispersion of nanofibrils, and indicates the presence of nanofibril clusters (Kaboarani and Riedl, 2012).

Fig. 7a and b show TEM images of bamboo nanofibrils after 5 and 30 passages through the defibrillator. Again, it is observed that 30 passages (Fig. 7b) improved the individualization of the nanofibrils in relation to 5 passages (Fig. 7a). Fig. 7c and d present the diameter distribution (measured in the TEM images) of the nanofibrils. The values measured after 5 and 30 passages were 123.1 ± 14.7 nm and 19.3 ± 9.2 nm, respectively, which were higher than those measured in the AFM images. Similar behavior was found in literature (Flauzino Neto et al., 2013). Results of the morphology obtained in the present work are consistent with nanofibers extracted from other sources (Karimi et al., 2014; Chen et al., 2011; Li et al., 2014; Abe and Yano, 2010; Tonoli et al., 2012).

3.3. Characterization of the nanocomposites

3.3.1. Mechanical properties

Fig. 8 shows the average and standard deviation values of tensile strength, elongation at break and tensile modulus of the blend (P80/A20) and nanocomposites reinforced with different contents of nanofibrils obtained with 5 (Fig. 8a–c) and 30 (Fig. 8d–f) passages through the defibrillator. Tensile strength, elongation at break and tensile modulus of the control blend (P80/A20) were 26.3 MPa, 186% and 503 MPa, respectively. No significant ($p \leq 0.05$) increase in tensile strength was observed for inclusion of bamboo nanofibrils obtained with 5 passages (Fig. 8a). Nanocomposites reinforced with 4.5% of nanofibrils from 5 passages presented significant ($p \leq 0.05$) higher values of elongation at break than the other samples, and

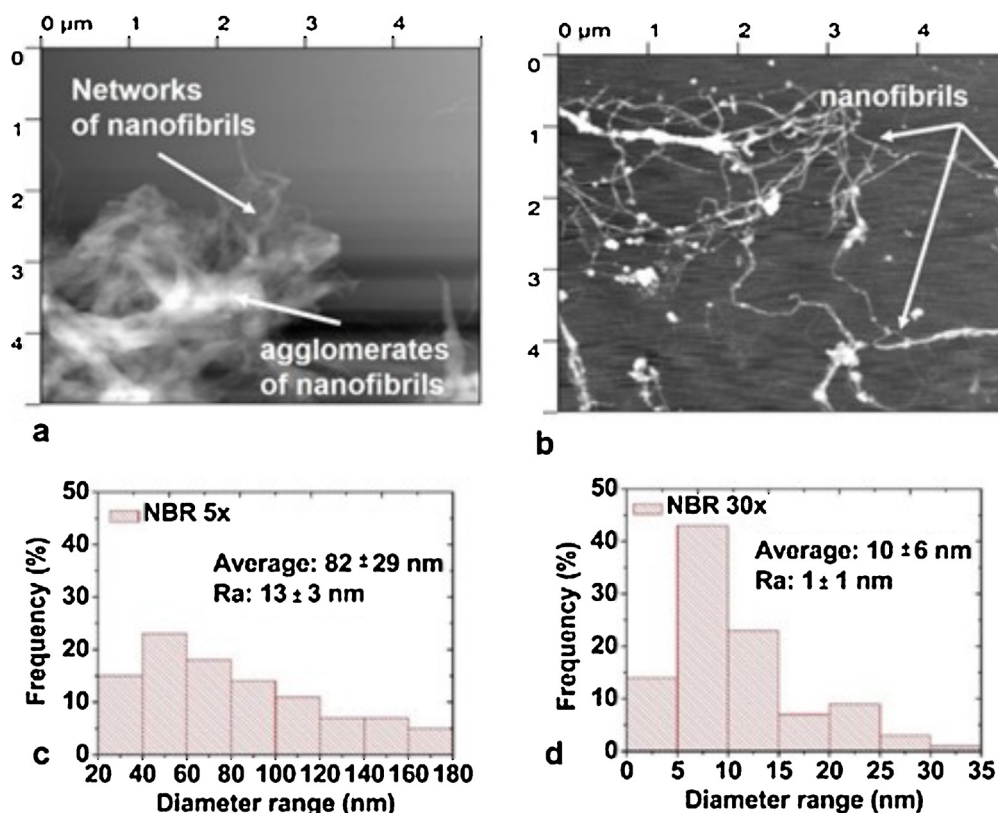


Fig. 6. Typical atomic force microscopy (AFM) images (topography) of the bamboo nanofibrils: (a) 5 passages (Guimarães Junior et al., 2015); and (b) 30 passages through the defibrillator. Diameter distribution (determined from measurements in the AFM images) of the bamboo nanofibrils: (c) 5 passages (Guimarães Junior et al., 2015); and (d) 30 passages through the defibrillator (Guimarães Junior et al., 2015). Average diameter of the nanofibrils and roughness (Ra) provided with the nanofibrils is presented. Scanning area was 25 μm² (5 μm × 5 μm).

leading to around 29% more ductile film than the control blend (Fig. 8b). Tensile modulus increased significantly (12% in relation to the control blend) for nanocomposite with 0.5% nanofibrils and decreased significantly for nanocomposites with 4.5% and 6.5% of nanofibrils (Fig. 8c).

The addition of 4.5 and 6.5% of bamboo nanofibrils obtained with 30 passages resulted on significant ($p \leq 0.05$) improvements of tensile strength (Fig. 8d) and elongation at break (Fig. 8e) of the nanocomposites. For 6.5% of nanofibrils the increase was of around 24% for tensile strength and 51% for elongation at break in relation to control blend. Tensile modulus decreased significantly (around 40%) in relation to the control blend (Fig. 8f).

The lower performance of the nanocomposites with nanofibrils obtained with 5 passages in relation to 30 passages is probably because the former presents some agglomeration of nanofibrils that may cause fissures and/or cracks in the microstructure of the nanocomposites. On the other hand, for nanocomposites reinforced with nanofibrils from 30 passages through the defibrillator, the large surface area of the smaller nanofibrils permitted greater efficiency in the load transference from the matrix to the nanofibrils (Bilbao-Sainz et al., 2011). The improved distribution of the nanofibrils led to more efficient packaging and compact structure. Similar results were found elsewhere (Petersson et al., 2007; Lu et al., 2008; Jonoobi et al., 2010; Xu et al., 2013).

The values of tensile strength obtained in the present work are similar to those of linear low density polyethylene – LLDPE (37 MPa), polycaprolactone – PCL (20.7–42 MPa), and polypropylene – PP (35 MPa); and higher than values presented by low density polyethylene – LDPE (6.9–16.0 MPa), polyesteramide – PEA (17 MPa) and highly branched LDPE (8.5–10.5 MPa); which are

polymers frequently used in flexible packages for food industry and agricultural sector (Doak, 1986; Averous, 2004; Auras et al., 2004).

3.3.2. Morphology of the blend and nanocomposite surfaces

SEM analyses permits visualize the homogeneity of the surface microstructure, including the presence of fissures, cracks and delaminations that may affect mechanical and barrier properties of the material. Fig. 9a and b present SEM micrographs of the control blend, which presented continuous and cohesive surface. No separations of the polymer phases (starch and PVA) were observed, demonstrating good interactions between them. Previous studies (Frost et al., 2011; Han et al., 2009; Tudorachi et al., 2000) also showed good compatibility between cassava starch and PVA due to the high presence of hydrogen bonding energy in both polymers.

Nanocomposites with 0.5% nanofibrils obtained after 5 passages presented some cracks and erupted bubbles (Fig. 9c). Similar results were reported by Jayasekara et al. (2004) and Rindlav-Westling et al. (2002), whose fissures may have been caused by surface wrinkling due to the drying process or differences in the drying shrinkage related to the presence of the nanofibrils and agglomerates. Specimens with 1.5 and 6.5% of nanofibrils obtained with 5 passages also presented great roughness due to the formation of agglomerates/clusters of nanofibrils (Fig. 9d and f). Among all nanocomposites reinforced with nanofibrils obtained with 5 passages, those with 4.5% concentration presented the best distribution within the blend matrix (Fig. 9e). It agrees with the higher average values of tensile strength (Fig. 8a) and elongation (Fig. 8b) for these nanocomposites in the previous section.

Fig. 10 depicts SEM micrographs of nanocomposites reinforced with nanofibrils obtained with 30 passages through the defibril-

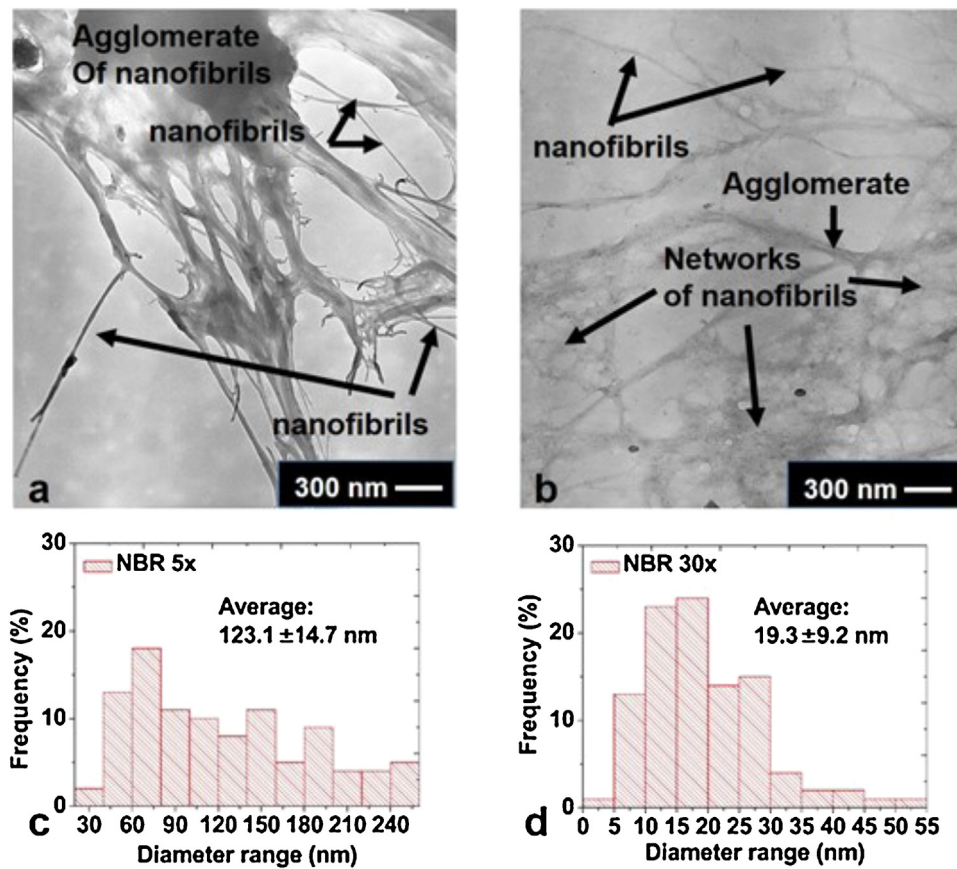


Fig. 7. Typical transmission electron microscopy (TEM) images of bamboo nanofibrils obtained with: (a) 5 passages; and (b) 30 passages through the defibrillator. Diameter distribution (determined from measurements in the TEM images) of the bamboo nanofibrils: (c) 5 passages; and (d) 30 passages.

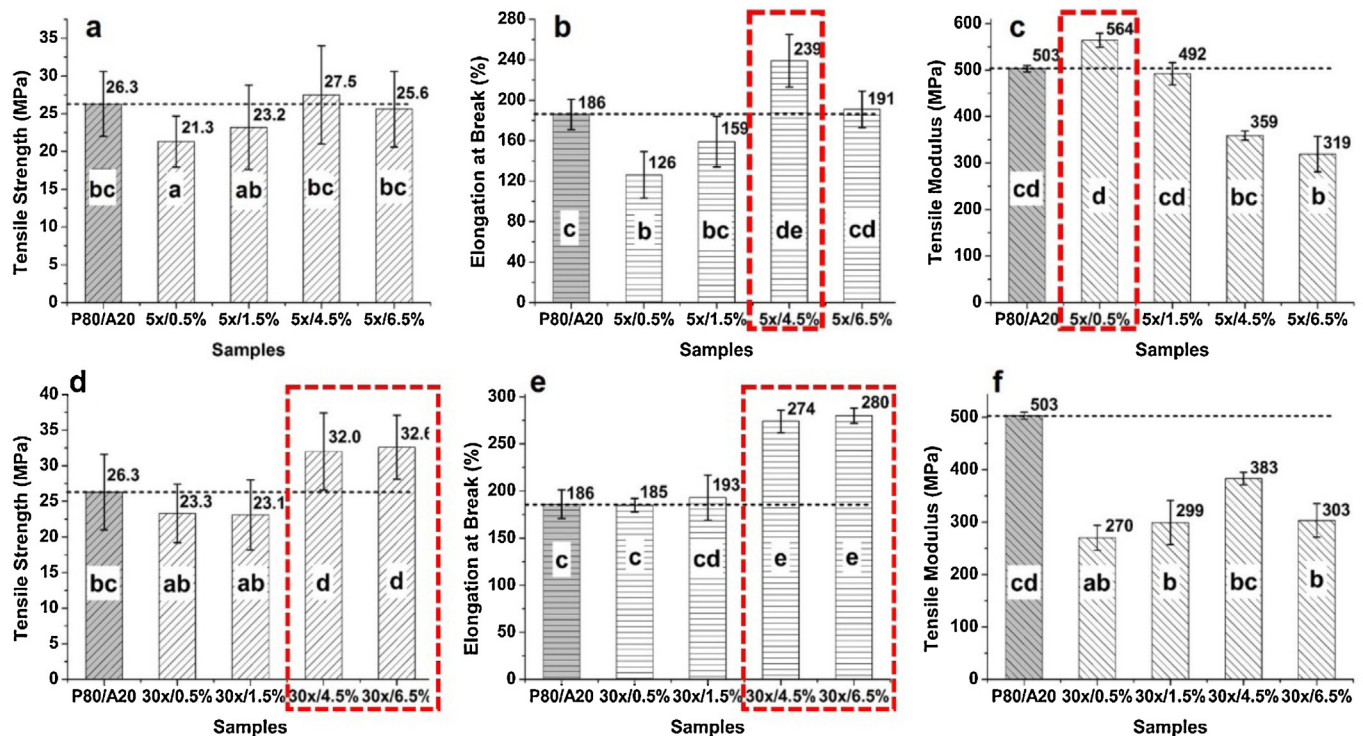


Fig. 8. Average and standard deviation values of: (a) and (d) tensile strength; (b) and (e) elongation at break; and (c) and (f) tensile modulus of the control blend (P80/A20) and nanocomposites reinforced with different contents (0.5%, 1.5%, 4.5% and 6.5% dry basis) of bamboo nanofibrils obtained with 5 passages (a–c) and 30 passages (d–f) through the defibrillator. Dotted lines indicate the average values of the reference material (control blend: P80/A20). Lower case letters in the bars (a, b, c, d and e) represent statistical comparisons using Fisher's test, while different letters indicate significant ($p \leq 0.05$) differences between the films.

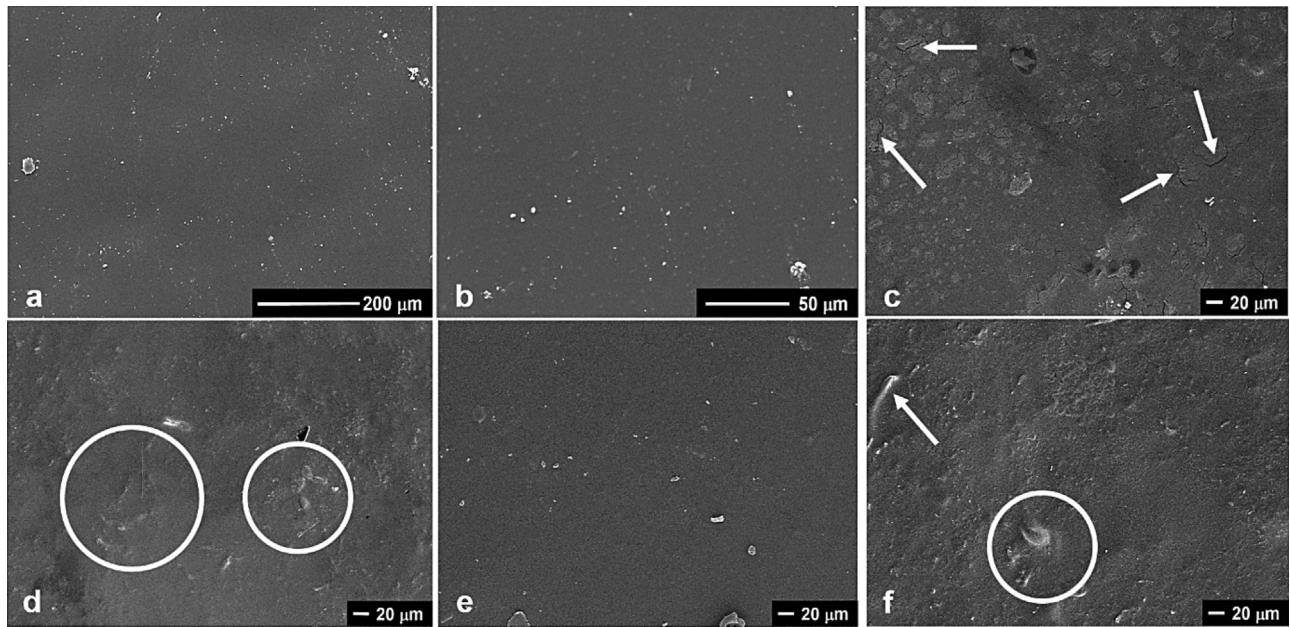


Fig. 9. Typical scanning electron microscopy (SEM) micrographs of: (a) and (b) control blends (P80/A20–80% of PVA and 20% of starch) with magnification of 300 \times and 1000 \times , respectively; and nanocomposites reinforced with different contents of bamboo nanofibrils obtained with 5 passages through the defibrillator: (c) 0.5%; (d) 1.5%, (e) 4.5% and (f) 6.5%, with magnification of 500 \times . Arrows and circles indicate fissures and the formation of clusters.

lator. In this case nanofibrils are better dispersed in the matrix for all concentrations. The more compact surface, the cohesiveness, the homogeneity of the film, the absence of phase separation and minimum presence of bubbles, clusters and fissures are good

evidences of the improved microstructure of this nanocomposites (Mali et al., 2004). As reported in the previous section, with 30 passages through the defibrillator, the nanofibrils showed a better interaction with the blend matrix due to the reduction of nanofibril

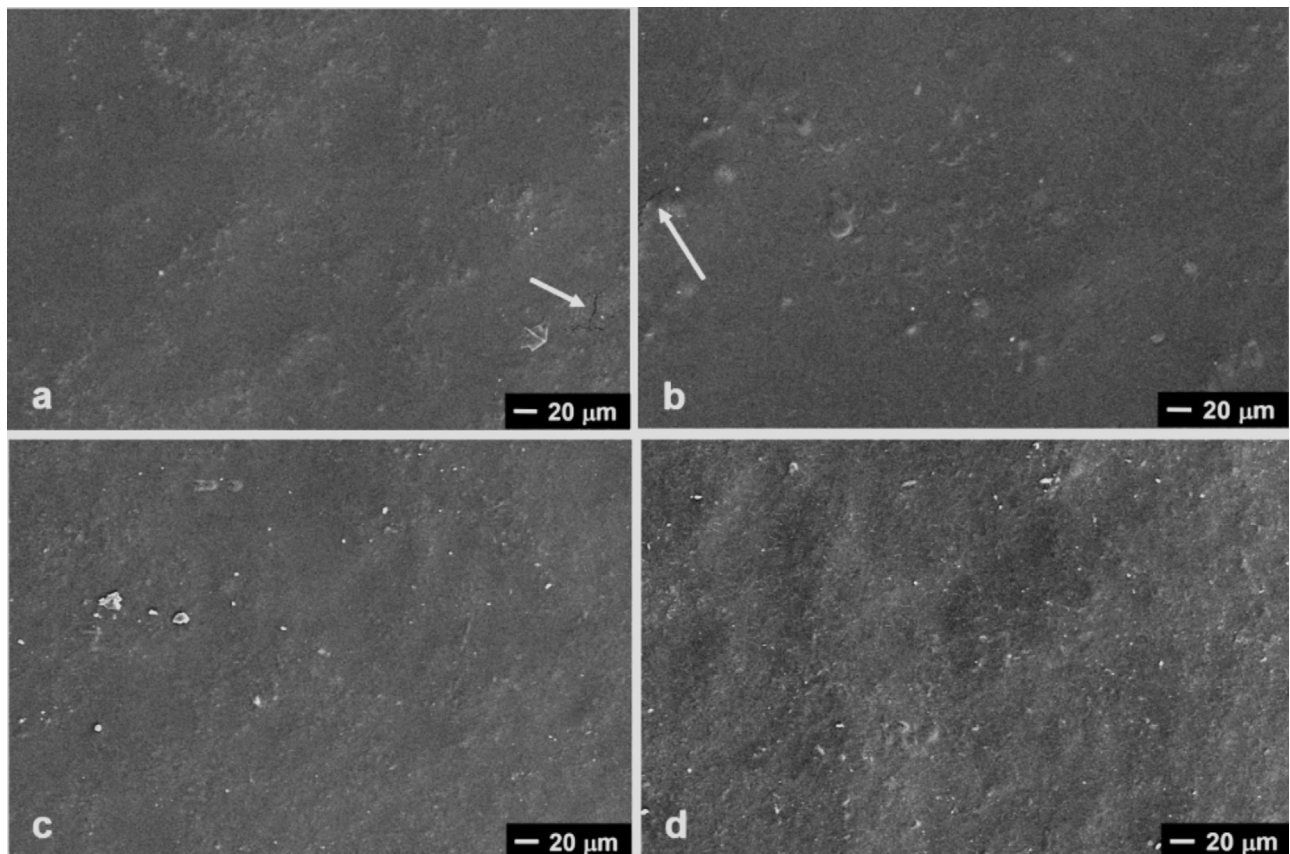


Fig. 10. Typical scanning electron microscopy (SEM) micrographs of nanocomposites reinforced with different contents of bamboo nanofibrils obtained with 30 passages through the defibrillator: (a) 0.5%; (b) 1.5%; (c) 4.5%; and (d) 6.5%. Arrows show fissures. Magnification: 500 \times .

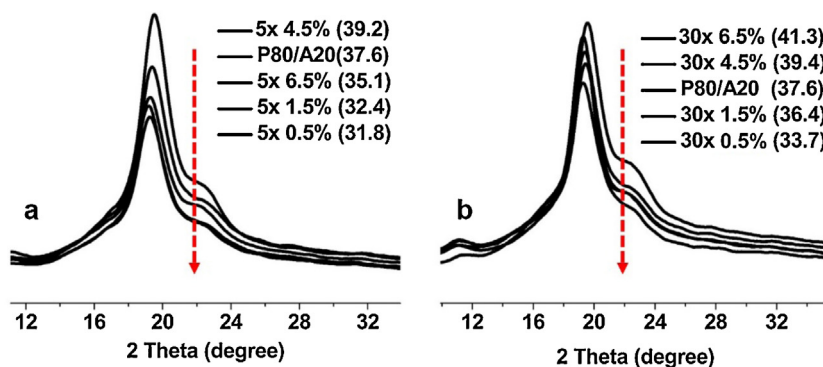


Fig. 11. X-ray diffraction (XRD) pattern of nanocomposites reinforced with: (a) bamboo nanofibrils in different contents (0.5%, 1.5%, 4.5% and 6.5% dry basis) obtained with 5 passages through the defibrillator; and (b) bamboo nanofibrils in different contents obtained with 30 passages. P80/A20 refers to the control blend of PVA and starch. Arrow indicates the order of the samples (as presented in the legends), while values in parentheses are the calculated crystalline indexes of the films.

diameter that improves surface contact and increase free hydroxyls groups for bonding. Again, this result is consistent with the mechanical performance.

3.3.3. X-ray diffractometry – DRX

In general, starch presents some crystalline structure attributed to well organized amylopectin molecules (Moorthy, 2002). PVA is typically semicrystalline with an amorphous band and some degree of crystallinity depending on its degree of hydrolysis (Bassner and Klingenberg, 1998). Fig. 11 shows the characteristic diffraction peak of PVA at $2\theta = 19.6^\circ$ (Das et al., 2010), and a shoulder at around $2\theta = 22.5^\circ$, which is assigned to the (002) lattice plane of cellulose I in bamboo nanofibrils and to the crystal structures of the starch. Fig. 11b shows that bamboo nanofibrils obtained with 30 passages increased the crystallinity index of the nanocomposites in relation to those reinforced with nanofibrils obtained with 5 passages (Fig. 11a). This is because well-dispersed nanofibrils probably acted as nucleating agents for the crystals (Fujisawa et al., 2014; Wang et al., 2014).

3.3.4. Physical properties

3.3.4.1. Optical transmittance (Tr). Fig. 12 shows the Tr curves for the blend and nanocomposites. Blends are more transparent ($Tr = 85\%$ at 800 nm wavelength) than nanocomposites. Nanocomposites reinforced with bamboo nanofibrils obtained with 30 passages presented higher values of transmittance (Tr between 79% and 84% at 800 nm wavelength) in relation to 5 passages, for all contents of nanofibrils (Fig. 12b). This is because nanofibrils from 30 passages are well dispersed and presents lower diameters, improving transparency of the films. Nanocomposites reinforced

with nanofibrils from 5 passages (Fig. 12a) presented lower transmittance values (Tr from 60% to 78%) probably due to the nanofibrils with high diameters and occasionally due to the poor dispersion of them.

Nanocomposites became more opaque as the concentration of nanofibrils increased for both passage numbers (5 \times and 30 \times), because of the formation of a percolated web and some nanofibril clusters that block the transmittance of the light. The SEM images corroborate this discussion (Figs. 9c–f and Fig. 10a–d). Similar results were presented elsewhere (Bilbao-Sainz et al., 2011; Savadekar and Mhaske, 2012; Iwamoto et al., 2005; Bielecki et al., 2005; Chávez-Pacheco et al., 2004). The optical properties are an important issue for packaging films, because may influence the appearance of the products. Normally, a transparent packaging is desirable in order to allow of viewing the packed product (Brawn, 1992). In other cases, it is necessary to protect the product from light in order to reduce catalyzed deteriorations (Sarantópoulos et al., 2002).

3.3.4.2. Water vapor permeability (WVP). The average values of water vapor transmission rate (WVTR) and water vapor permeability (WVP) for nanocomposites reinforced with nanofibrils obtained with 5 and 30 passages are presented in Table 1. Lower contents (0.5% and 1.5%) of nanofibrils led to some increase of WVTR and WVP probably because they are not enough to form a consistent network of nanofibrils capable of retain the penetration of water vapor. Instead, this low content of nanofibrils forms clusters and some porosity that causes a negative effect on barrier properties (Sanchez-Garcia et al., 2007). SEM images presented in previous sections (Fig. 9c–d and Fig. 10a–b) corroborate with

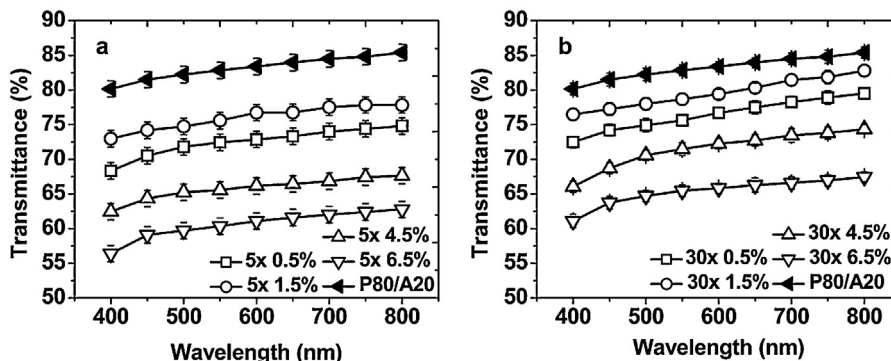


Fig. 12. Average and standard-deviation values of light transmittance (Tr) in the range of 400–800 nm wavelength, through the blend and nanocomposites: (a) containing bamboo nanofibrils (in different contents) obtained with 5 passages through the defibrillator; (b) obtained with 30 passages. The filled triangles refer to the control blend (P80/A20, composed of 80% PVA and 20% cassava starch).

Table 1

Average and standard deviations values of water vapor transmission rate (WVTR) and water vapor permeability (WVP) of the control blend (P80/A20) and nanocomposites reinforced with different concentrations of bamboo nanofibrils obtained with 5 and 30 passages through the defibrillator.

Material	Nanofibrils ^a (%)	Thickness ^b (mm)	WVTR ^c (gH ₂ O/m ² day)	WVP ^c (g mm/KPa day m ²)
Nanocomposite with nanofibrils from 5 passages	0.5	0.085 ± 0.009	33.68 ± 0.61	1.31 ± 0.02 d
	1.5	0.099 ± 0.007	30.34 ± 0.95	1.26 ± 0.01 c
	4.5	0.143 ± 0.012	24.60 ± 0.97	1.07 ± 0.02 a
	6.5	0.118 ± 0.011	27.16 ± 1.57	1.16 ± 0.03 b
Nanocomposite with nanofibrils from 30 passages	0.5	0.081 ± 0.006	31.54 ± 0.64	1.25 ± 0.02 b
	1.5	0.099 ± 0.008	29.82 ± 1.18	1.23 ± 0.01 b
	4.5	0.127 ± 0.011	21.16 ± 0.44	0.98 ± 0.03 a
	6.5	0.105 ± 0.009	20.67 ± 1.09	0.94 ± 0.02 a
Control blend (P80/A20) ^d	–	0.077 ± 0.002	29.13 ± 0.22	1.18 ± 0.01 b

^a Content in dry weight.

^b Average of 10 measurements.

^c Average of 3 measurements.

^d Blend composed of 80% PVA and 20% cassava starch. Lower case letters (a–d) in the SOL column represent statistical comparisons using Scott–Knott test, while different letters indicate significant ($p \leq 0.05$) differences between the values.

Table 2

Average and standard deviations values of water solubility (SOL) for the control blend (P80/A20) and nanocomposites reinforced with different concentrations of bamboo nanofibrils obtained with 5 and 30 passages through the defibrillator.

Material	Nanofibrils ^a (%)	Thickness ^b (mm)	SOL ^c (%)
Nanocomposite with nanofibrils from 5 passages	0.5	0.0803 ± 0.012	30.08 ± 8.4 c
	1.5	0.0887 ± 0.011	28.33 ± 9.7 b
	4.5	0.1231 ± 0.012	25.82 ± 10.8 a
	6.5	0.1064 ± 0.014	25.66 ± 9.8 a
Nanocomposite with nanofibrils from 30 passages	0.5	0.0776 ± 0.009	28.54 ± 7.7 d
	1.5	0.0883 ± 0.012	25.82 ± 9.1 c
	4.5	0.1131 ± 0.009	21.16 ± 7.7 b
	6.5	0.1077 ± 0.011	17.33 ± 9.1 a
Control blend (P80/A20) ^d	–	0.0682 ± 0.004	25.47 ± 4.6 c

^a Content in dry weight.

^b Average of 10 measurements.

^c Average of 3 measurements.

^d Blend composed of 80% PVA and 20% cassava starch. Lower case letters (a–d) in the SOL column represent statistical comparisons using Scott–Knott test, while different letters indicate significant ($p \leq 0.05$) differences between the values.

this results, showing the formation of pores, and fissures/cracks on those nanocomposite surfaces.

Addition of high contents of nanofibrils (4.5% and 6.5%) from 5 and 30 passages decreased the WVTR and WVP values of the nanocomposites in relation to the control blend (P80/A20). This decrease in permeability may be explained by the percolated network formed by the nanofibers, leading to a continuous network of hydrogen bonds that reduces the diffusion of water vapor (Sreekala et al., 2008; Lai and Padua, 1998; Kaushik et al., 2010). Similar behavior was found elsewhere (Mikkonen et al., 2011; Svagan et al., 2009; Dufresne et al., 2000). The values of WVTR observed in the present work (20–30 g/m².day) are higher than values found for some polymers used in the packaging sector (low-density polyethylene – LDPE = 3.7 g/m².day, high-density polyethylene – HDPE = 3.5 g/m².day) and polylactic acid – PLA (12.6 g/m².day) (Petersen et al., 2001).

3.3.4.3. Water solubility (SOL). The control blend and all nanocomposites evaluated became opaque after 24 h immersed in water, while continuing flexible and foldable. Water solubility was significantly higher ($p > 0.05$) for nanocomposites with 0.5% nanofibrils (in both number of passages) (Table 2). Water solubility was around 30% lower (in comparison to the control blend) for nanocomposites reinforced with 6.5% of nanofibrils obtained with 30 passages through the defibrillator. As discussed before, this result is consequence of the lower diameters of the nanofibrils produced in this condition, which improved the percolated network (Müller et al., 2009) and decreased pores and fissures in these nanocomposites. In contrast, the formation of fissures and pores occurred

in specimens with fibril agglomerations or insufficient amount of nanofibrils bonded together. Results of water solubility agree with the findings for vapor permeability.

4. Conclusions

Bamboo nanofibrils were effectively isolated from commercial refined bamboo pulp. Chemical pre-treatments (alkaline and bleaching) were used to improve the individualization of the nanofibrils in the mechanical defibrillator. Morphological analysis using atomic force microscopy (AFM) indicated a reduction of around 88% (from 82 ± 29 nm to 10 ± 6 nm) in the average diameter of the nanofibrils obtained with 30 passages through the defibrillator, which made the nanofibril suspensions more stable in relation to 5 passages. Tensile strength and elongation at breaking of the nanocomposite films were increased in up to 24 and 51%, respectively, for the high contents (4.5 and 6.5) of nanofibrils compared to the control blend. Light transmission of the nanocomposite films were better for nanofibrils obtained with 30 passages, and decreased with the increase on the nanofibril content. Results showed that the higher contents (4.5% and 6.5%) of nanofibrils reduced up to 20 and 30% the values of water vapor permeability and water solubility, respectively, in relation to the control blend. The present work demonstrates the feasibility of producing bamboo nanofibrils (by mechanical defibrillation) that can be used as reinforcement to engineer polymeric nanocomposites for a myriad of applications.

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References

- Abe, K., Yano, H., 2010. Comparison of the characteristics of cellulose microfibril aggregates isolated from fiber and parenchyma cells of Moso bamboo (*Phyllostachys pubescens*). *Cellulose* 17, 271–277, <http://dx.doi.org/10.1007/s10570-009-9382-1>.
- American Society for Testing and Materials, 2000. Standard test methods for tensile properties of thin plastic sheeting. In: ASTM D882-00, Philadelphia.
- American Society for Testing and Materials, 2000. Standard test methods for water vapor transmission of materials. In: ASTM E96-00, Philadelphia.
- Auras, R., Harte, B., Selke, S., 2004. An overview of polylactides as packaging materials. *Macromol. Biosci.* 4, 835–864.
- Averous, L., 2004. Biodegradable multiphase system based on plasticized starch: a review. *J. Macromol. Sci. Part C-Polym. Rev.* 44, 231–274.
- Averous, L., Boquillon, N., 2004. Biocomposites based on plasticized starch: thermal and mechanical behaviours. *Carbohydr. Polym.* 56, 111–122.
- Samir, Azizi, Alloin, M.A., F. Dufresne, A., 2005. Review of recent research into cellulosic whiskers their properties and their application in nanocomposite field. *Biomacromolecules* 6, 612–626, <http://dx.doi.org/10.1021/bm0493685>.
- Bassner, S.L., Klingenberg, E.H., 1998. Using poly(vinyl alcohol) as a binder. *Am. Ceram. Soc. Bull.* 77, 71–75.
- Beliakova, M.K., Aly, A.A., Abdel-Mohdy, F.A., 2004. Grafting of poly(methacrylic acid) on starch and poly(vinyl alcohol). *Starch-Stärke* 56, 407–412.
- Bielecki, S., Krystynowicz, A., Turkiewicz, M., Kalinowska, H., 2005. Bacterial cellulose. *Biopolym. Online* 14, 381, <http://dx.doi.org/10.1002/3527600035.bpol5003>.
- Bilbao-Sainz, C., Bras, J., Williams, T., Sénéchal, T., Orts, W., 2011. HPMC reinforced with different cellulose nano-particles. *Carbohydr. Polym.* 86, 1549–1557, <http://dx.doi.org/10.1016/j.carbpol.2011.06.060>.
- Brawn, W.E., 1992. *Plastics in Food Packaging Properties, Design and Fabrication*, 8–10. Marcel Dekker Inc., New York, pp. 539.
- Buafino, L., Mendes, L.M., Tonoli, G.H.D., Rodrigues, A., Fonseca, A.S., Cunha, P.I., Marconcini, J.M., 2014. New products made with lignocellulosic nanofibers from Brazilian amazon forest. *IOP Conf. Ser. Mater. Sci. Eng.* 64, 012012, <http://dx.doi.org/10.1088/1757-899X/64/1/012012>.
- Chang, P.R., Jian, R., Yu, J., Ma, X., 2010. Starch-based composites reinforced with novel chitin nanoparticles. *Carbohydr. Polym.* 80, 421–426.
- Chávez-Pacheco, J.L., Yee, S.M., Zentella, M.C., Marván, E.E., 2004. Celulosa bacteriana en gluconacetobacter xylinum: biosíntesis y aplicaciones. *Rev. Especial. Cienc. Quím.-Biol.* 7, 18–25.
- Chen, H.Z., Chen, J.Z., Liu, J., Li, Z.H., 1999. Studies on the steam explosion of wheat straw: effects of the processing conditions for steam explosion of wheat straw and analysis of the process. *J. Cellul. Sci. Technol.* 7, 60–67.
- Chen, Y., Cao, X., Chang, P.R., Huneault, M.A., 2008. Comparative study on the films of poly(vinyl alcohol)/pea starch nanocrystals and poly(vinyl alcohol)/native pea starch. *Carbohydr. Polym.* 73, 8–17, <http://dx.doi.org/10.1016/j.carbpol.2007.10.015>.
- Cinelli, P., Chiellini, E., Lawton, J.W., Imam, S.H., 2006. Foamed articles based on potato starch corn fibers and poly(vinyl alcohol). *Polym. Degrad. Stab.* 91, 1147–1155.
- Corradini, E., Morais, L.C., Rosa, M.F., Mazzetto, S.E., Mattoso, L.H., Agnelli, J.A.M.A., 2006. Preliminary study for the use of material natural fibres as reinforcement in starch–gluten–glycerol matrix. *Macromol. Symp.* 245–246, 558–564.
- Das, K., Ray, D., Bandyopadhyay, N.R., Gupta, A., Sengupta, S., Sahoo, S., 2010. Preparation and characterization of cross-linked starch/poly(vinyl alcohol) green films with low moisture absorption. *Ind. Eng. Chem. Res.* 49, 2176–2185, <http://dx.doi.org/10.1021/ie901092n>.
- Dinand, E., Chanzy, H., Vignon, M.R., 1999. Suspensions of cellulose microfibrils from sugar beet pulp. *Food Hydrocolloids* 13, 275–283, [http://dx.doi.org/10.1016/S0268-005X\(98\)00084-8](http://dx.doi.org/10.1016/S0268-005X(98)00084-8).
- Doak, K.W., 1986. Ethylene Polymers. In: Mark, H.M., Bilakes, N.M., Overberg, C.G., Mendes, G. (Eds.), *Encyclopedia of Polymer Science and Engineering*, 6. John-Wiley & Sons, New York.
- Dufresne, A., Cavaille, J.Y., Vignon, M.R., 1997. Mechanical behavior of sheets prepared from sugar beet cellulose microfibrils. *J. Appl. Polym. Sci.* 64, 1185–1194.
- Dufresne, A., Dupeyre, D., Vignon, M.R., 2000. Cellulose microfibrils from potato tuber cells: processing and characterization of starch–cellulose microfibril composites. *J. Appl. Polym. Sci.* 76, 2080–2092, [http://dx.doi.org/10.1002/\(SICI\)1097-4628\(20000628\)76:14<2080::AID-APP12>3.0.CO;2-U](http://dx.doi.org/10.1002/(SICI)1097-4628(20000628)76:14<2080::AID-APP12>3.0.CO;2-U).
- Flauzino Neto, W.P., Silvério, H.A., Dantas, N.O., Pasquini, D., 2013. Extraction and characterization of cellulose nanocrystals from agro-industrial residue–Soy hulls. *Ind. Crops Prod.* 42, 480–488, <http://dx.doi.org/10.1016/j.indcrop.2012.06.041>.
- Follain, N., Joly, C., Dole, P., Bliard, C., 2005. Mechanical properties of starch-based materials: I. short review and complementary experimental analysis. *J. Appl. Polym. Sci.* 97, 1783–1794.
- Frost, K., Barthes, J., Kaminski, D., Lascaris, E., Niere, J., Shanks, R., 2011. Thermoplastic starch–silica–polyvinyl alcohol composites by reactive extrusion. *Carbohydr. Polym.* 84, 343–350.
- Fujisawa, S., Zhang, J., Saito, T., Iwata, T., Isogai, A., 2014. Cellulose nanofibrils as templates for the design of poly(L-lactide) nucleating surfaces. *Polymer* 55, 2937–2942.
- Gomide, J.L., 1986. Estudo sobre a constituição química do *Bambusa vulgaris*, visando a produção de polpa celulósica. *O Papel* 47, 64–68.
- Gomide, J.L., Colodette, J.L., Oliveira, R.C., 1982. Estudo das possibilidades do *Bambusa vulgaris* para produção de papéis tipo kraft. *O Papel* 28, 38–42.
- Gontard, N., Duche, C., Cuq, J.L., Guilbert, S., 1994. Edible composite films of wheat gluten and lipids: water vapour permeability and other physical properties. *Int. J. Food Sci. Technol.* 29, 39–50, <http://dx.doi.org/10.1111/j.1365-2621.1994.tb02045.x>.
- Guimarães Junior, M., Botaro, V.R., Novack, K.M., Flauzino Neto, W.P., Mendes, L.M., Tonoli, G.H.D., 2015. Preparation of cellulose nanofibrils from bamboo pulp by mechanical defibrillation for their applications in biodegradable composites. *J. Nanosci. Nanotechnol.* 15, <http://dx.doi.org/10.1166/jnn.2015.10854> (in press).
- Han, X.Z., Chen, S.S., Hu, X.G., 2009. Controlled-release fertilizer encapsulated by starch/polyvinyl alcohol coating. *Desalination* 240, 21–26.
- Ifuku, S., Nogi, M., Yoshioka, M., Morimoto, M., Yano, H., 2010. Fibrillation of dried chitin into 10–20 nm nanofibers by a simple grinding method under acidic condition. *Carbohydr. Polym.* 81, 134–139.
- Ishiaku, U.S., Pang, K.W., Lee, W.S., Ishak, Z.A., 2002. Mechanical properties and enzymic degradation of thermoplastic and granular sago starch filled poly(caprolactone). *Eur. Polym. J.* 38, 393–401.
- Iwamoto, S., Nakagaito, A.N., Yano, H., Nogi, M., 2005. Optically transparent composites reinforced with plant fiber-based nanofibers. *Appl. Phys. A* 81, 1109–1112, <http://dx.doi.org/10.1007/s00339-005-3316-z>.
- Jayasekara, R., Harding, I., Bowater, I., Christie, G.B.Y., Lonergan, G.T., 2004. Preparation, surface modification and characterisation of solution cast starch PVA blended films. *Polym. Test.* 23, 17–27.
- Jonoobi, M., Harun, J., Mathew, A.P., Oksman, K., 2010. Mechanical properties of cellulose nanofiber (CNF) reinforced polylactic acid (PLA) prepared by twin screw extrusion. *Compos. Sci. Technol.* 70, 1742–1747.
- A. Kaboorani, B. Riedl, In: Blanchet quest editor. Pierre: Felin M, Hosseinaei O, Wang S. Nanocrystalline cellulose (NCC): a renewable nano-material for polyvinyl acetate (PVA) adhesive, *Euro Polym J* 48 2012; 1829–1837.
- Kampeerappun, P., Aht-Ong, D., Pentrakoon, D., Srikulkit, K., 2007. Preparation of cassava starch/montmorillonite composite film. *Carbohydr. Polym.* 67, 155–163, <http://dx.doi.org/10.1016/j.carbpol.2006.05.012>.
- Karimi, S., Tahir, P., Karimi, A., Dufresne, A., Abdulkhani, A., 2014. Kenaf bast cellulosic fibers hierarchy: a comprehensive approach from micro to nano. *Carbohydr. Polym.* 101, 878–885.
- Kaushik, A., Singh, M., Verma, G., 2010. Green nanocomposites based on thermoplastic starch and steam exploded cellulose nanofibrils from wheat straw. *Carbohydr. Polym.* 82, 337–345, <http://dx.doi.org/10.1016/j.carbpol.2010.04.063>.
- Khan, M.A., Bhattacharia, S.K., Kader, M.A., Bahari, K., 2006. Preparation and characterization of ultra violet (UV) radiation cured bio-degradable films of sago starch/PVA blend. *Carbohydr. Polym.* 63, 500–506.
- Kvien, L., Tanem, B.S., Oksman, K., 2005. Characterization of cellulose whiskers and their nanocomposites by atomic force and electron microscopy. *Biomacromolecules* 6, 3160–3165, <http://dx.doi.org/10.1021/bm050479t>.
- Lai, H.M., Padua, G.W., 1998. Water vapor barrier properties of zein films plasticized with oleic acid. *Cereal Chem.* 75, 194–199, <http://dx.doi.org/10.1094/CCEM.1998.75.2.194>.
- Lawton, J.W., 1996. Effect of starch type on the properties of starch containing films. *Carbohydr. Polym.* 29, 203–208.
- Li, M., Wang, L.J., Li, D., Cheng, Y.L., Adhikar, B., 2014. Preparation and characterization of cellulose nanofibers from depectinated sugar beet pulp. *Carbohydr. Polym.* 102, 1346–1403.
- Li, X.B., Shape, T.F., Peter, G.F., Hse, C.Y., Eberhardt, T.L., 2007. Chemical changes with maturation of the bamboo species *Phyllostachys pubescens*. *J. Trop. Forest Sci.* 19, 6–12.
- Lu, J., Wang, T., Drzal, L.T., 2008. Preparation and properties of microfibrillated cellulose polyvinyl alcohol composite materials. *Compos. Part A Appl. Sci. Manuf.* 39, 738–746, <http://dx.doi.org/10.1016/j.compositesa.2008.02.003>.
- Mali, S., Karam, L.B.R., Ramos, L.P., Grossmann, M.V.E., 2004. Relationships among the composition and physicochemical properties of starches with the characteristics of their films. *J. Agric. Food Chem.* 52, 7720–7725, <http://dx.doi.org/10.1021/jf049225+>.
- Mikkonen, K.S., Stevanic, J.S., Joly, C., Dole, P., Pirkkalainen, K., Serimaa, R., 2011. Composite films from spruce galactoglucomannans with microfibrillated spruce wood cellulose. *Cellulose* 18, 713–726.

- Mohanty, A.K., Misra, M., Hinrichsen, G., 2000. Biofibres, biodegradable, polymer and composites: an overview. *Macromol. Mater. Eng.* 276–277, 1–24.
- Moorthy, S.N., 2002. Physicochemical and functional properties of tropical tuber starches: a review. *Starch-Stärke* 54, 559–592, [http://dx.doi.org/10.1002/1521-379X\(200212\)54:12<559::AID-STAR222559>3.0.CO;2-2](http://dx.doi.org/10.1002/1521-379X(200212)54:12<559::AID-STAR222559>3.0.CO;2-2).
- R.L.C. Morais M., Benachour A.C., Duarte-Coelho Estudo da caracterização reológica do licor negro do processo soda/Bambusa vulgaris Schrad e o efeito da adição de peróxido de hidrogênio In: Congresso Iberoamericano de Investigación en celulosa y papel 2000.
- Müller, C.M.O., Laurindo, J.B., Yamashita, F., 2009. Effect of cellulose fibers addition on the mechanical properties and water vapor barrier of starch-based films. *Food Hydrocolloids* 23, 1328–1333, <http://dx.doi.org/10.1016/j.foodhyd.2008.09.002>.
- Mwaikambo, L.Y., 2006. Review of the history, properties and application of plant fibres. *Afr. J. Sci. Technol.* 7, 120–133.
- Nabar, Y.U., Draybuck, D., Narayan, R., 2006. Physicomechanical and hydrophobic properties of starch foams extruded with different biodegradable polymers. *J. Appl. Polym. Sci.* 102, 58–68.
- Nakagaito, A.N., Yano, H., 2004. The effect of morphological changes from pulp fiber towards nano-scale fibrillated cellulose on the mechanical properties of high-strength plant fiber based composites. *Appl. Phys. A* 78, 547–552.
- Nguyen, X.T., Tan, Z., 2009. Surface treatment with texturized microcrystalline cellulose microfibrils for improved paper and paper board, Patent United States Patent, 7497924.
- Okuyama, A., Motoki, M., Yamanaka, S., 1993. Bacterial cellulose IV. Application to processed foods. *Food Hydrocolloids* 6, 503–511, [http://dx.doi.org/10.1016/S0268-005X\(09\)80074-X](http://dx.doi.org/10.1016/S0268-005X(09)80074-X).
- Olobarrieta, I., 2005. Strategies to improve the aging, barrier and mechanical properties of chitosan whey and wheat gluten protein films. In: Tesis de doutorado. Department of Fibre and Polymer Technology. Royal Institute of Technology, Stockholm, Sweden.
- Orts, W.J., Shey, J., Imam, S.H., Glenn, G.M., Guttman, M.E., Revol, J.F., 2005. Application of cellulose microfibrils in polymer nanocomposites. *J. Polym. Environ.* 13, 301–306, <http://dx.doi.org/10.1007/s10924-005-5514-3>.
- Park, S.I., Zhao, Y., 2004. Incorporation of a high concentration of mineral or vitamin into chitosan-based films. *J. Agric. Food Chem.* 52, 1933–1939, <http://dx.doi.org/10.1021/jf034612p>.
- A.L.S. Pereira, D.M. Nascimento, J.P.S. Morais, M.S.M. Souza Filho, M.F. Rosa, Valorização de resíduos agroindustriais: uso do pseudocaule de bananeira como matéria-prima para obtenção de nanoestruturas de celulose. In: Encontro de Iniciação Científica da Embrapa Agroindústria Tropical, 8., 2010. Fortaleza. Resumos. Fortaleza: Embrapa Agroindústria Tropical, <http://ainfo.cnptia.embrapa.br/digital/bitstream/item/34438/1/RE10163.pdf.>.
- Petersen, K., Nielsen, P.V., Olsen, M.B., 2001. Physical and mechanical properties of biobased materials—starch polylactate and polyhydroxybutyrate. *Starch-Stärke* 53, 356–361.
- Petersson, L., Kvien, I., Oksman, K., 2007. Structure and thermal properties of poly(lactic acid)/cellulose whiskers nanocomposite materials. *Compos. Sci. Technol.* 67, 2535–2544.
- Ramaraj, B., 2007. Crosslinked poly(vinyl alcohol) and starch composite films.II. Physicomechanical: thermal properties and swelling studies. *J. Appl. Polym. Sci.* 103, 909–916.
- Rindlav-Westling, A., Stading, M., Gatenholm, P., 2002. Crystallinity and morphology in films of starch amylose and amylopectin blends. *Biomacromolecules* 3, 84–91, <http://dx.doi.org/10.1021/bm010114i>.
- Sanchez-Garcia, M.D., Gimenez, E., Lagaron, J.M., 2007. Morphology and barrier properties of solvent cast composites of thermoplastic biopolymers and purified cellulose fibers. *Carbohydr. Polym.* 71, 235–244, <http://dx.doi.org/10.1016/j.carbpol.2007.05.041>.
- Sarantopoulos, C.G.L., Oliveira, L.M., Padula, M., Coltro, L., Alves, R.M.V., Garcia, E.E.C., 2002. Embalagens plásticas flexíveis: principais polímeros e avaliação de propriedades. Campinas: CETEA/ITAL 1, 267.
- Savadekar, N.R., Mhaske, S.T., 2012. Synthesis of nano cellulose fibers and effect on thermoplastics starch based films. *Carbohydr. Polym.* 89, 146–151, <http://dx.doi.org/10.1016/j.carbpol.2012.02.063>.
- Siddaramaiah, S., Raj, B., Somashekar, R., 2004. Structure–property relation in polyvinyl alcohol/starch composites. *J. Appl. Polym. Sci.* 91, 630–635.
- Silvério, H.A., Flauzino Neto, W.P., Dantas, N.O., Pasquini, D., 2013. Extraction and characterization of cellulose nanocrystals from corn cob for application as reinforcing agent in nanocomposites. *Ind. Crops Prod.* 44, 427–436.
- Siqueira, G., Bras, J., Dufresne, A., 2010. Cellulosic bionanocomposites: a review of preparation, properties and applications. *Polymers* 2, 728–765.
- Siro, I., Plackett, D., 2010. Microfibrillated cellulose and new nanocomposite materials: a review. *Cellulose* 17, 459–494, <http://dx.doi.org/10.1007/s10570-010-9405-y>.
- Sreedhar, B., Chattopadhyay, D.K., Karunakar, M.S.H., Sastry, A.R.K., 2006. Thermal and surface characterization of plasticized starch polyvinyl alcohol blends crosslinked with epichlorohydrin. *J. Appl. Polym. Sci.* 101, 25–34.
- Sreekala, M.S., Goda, K., Devi, P.V., 2008. Sorption characteristics of water, oil and diesel in cellulose nanofiber reinforced corn starch resin/ramie fabric composites. *Compos. Interfaces* 15, 281–299.
- Suess, H.U., 2010. Pulp Bleaching Today. De Gruyter.
- Svagan, A.J., Hedenqvist, M.S., Berglund, L., 2009. Reduced water vapour sorption in cellulose nanocomposites with starch matrix. *Compos. Sci. Technol.* 69, 500–506.
- Syverud, K., Chinga-Carrasco, G., Toledo, J., Toledo, P.G., 2011. A comparative study of *Eucalyptus* and *Pinus Radiata* pulp fibres as raw materials for production of cellulose nanofibrils. *Carbohydr. Polym.* 84, 1033–1038, <http://dx.doi.org/10.1016/j.carbpol.2010.12.066>.
- TAPPI Useful Method Technicall Association of the Pulp and Paper Industry, 1999a: preparation of wood for chemical analysis (Test Method T 264cm-97). Atlanta, USA.
- TAPPI Useful Method Technicall Association of the Pulp and Paper 1999b: sampling and preparing wood for analysis (Test Method T 257cm-97). Atlanta, USA.
- TAPPI Useful Method Technicall Association of the Pulp and Paper 1999c: Holocellulose in wood (Test Method T 9 m- 54). Atlanta, USA.
- TAPPI Useful Method Technicall Association of the Pulp and Paper 1999d: alpha, beta and gamma cellulose in pulp (Test Method T 203cm-99). Atlanta, USA.
- TAPPI Useful Method. Technicall Association of the Pulp and Paper 1999e: acid insoluble lignin in wood and pulp (Test Method T 222cm-88). 1999. Atlanta, USA.
- TAPPI Useful Method Technicall Association of the Pulp and Paper 1999f: ash in wood, pulp, paper and paperboard: combustion at 525° C (Test Method T 211cm-93). Atlanta, USA.
- TAPPI Useful Method Technicall Association of the Pulp and Paper 1999g: solvent extractives of wood and pulp (Test Method T 204cm-97). Atlanta, USA.
- Tetens, V.O., 1930. Über einige meteorologische Begriffe. *Z. Geophysik* 6, 297–309.
- Tonoli, G.H.D., Teixeira, E.M., Correa, A.C., Marconcini, J.M., Caixeta, L.A., Pereira-da-Silva, M.A., Mattoso, L.H.C., 2012. Cellulose micro/nanofibers from eucalyptus kraft pulp: preparation and properties. *Carbohydr. Polym.* 89, 80–88.
- Tudorachi, N., Casaval, C.N., Rusu, M., Pruteanu, M., 2000. Testing of polyvinyl alcohol and starch mixtures as biodegradable polymeric materials. *Polym. Test.* 19, 785–799, [http://dx.doi.org/10.1016/S0142-9418\(99\)00049-5](http://dx.doi.org/10.1016/S0142-9418(99)00049-5).
- Utracki, L.A., 1989. Polymer and Blends: Thermodynamics and Rheology, 1st ed. Hanser Publishers, New York.
- Villanova, J.C.O., Ayres, E., Carvalho, S.M., Patrício, O.S., Pereira, F.V., Oréfice, R.L., 2011. Pharmaceutical acrylic beads obtained by suspension polymerization containing cellulose nanowhiskers as excipient for drug delivery. *Eur. J. Pharm. Sci.* 42, 406–415, <http://dx.doi.org/10.1016/j.ejps.2011.01.005>.
- Wahab, R., Mustafa, M.T., Salam, of four cultivated tropical bamboo in genus *Gigantochloa*. *J. Agric. Sci.* 5, 66–75, <http://dx.doi.org/10.5539/jas.v5n8p66>.
- Wang, X., Sun, H., Bai, H., Zhang, L.P., 2014. Thermal, mechanical, and degradation properties of nanocomposites prepared using lignin-cellulose nanofibers and poly (lactic acid). *BioResource* 9, 3211–3224.
- Xu, X., Liu, F., Jiang, L., Zhu, J.Y., Haagensohn, D., Wiesenborn, D.P., 2013. Cellulose nanocrystals vs. cellulose nanofibrils: a comparative study on their microstructures and effects as polymer reinforcing agents. *ACS Appl. Mater. Interfaces* 5, 2999–3009, <http://dx.doi.org/10.1021/am302624t>.
- Zhai, M.L., Yoshii, F., Kume, T., 2003. Radiation modification of starch-based plastic sheets. *Carbohydr. Polym.* 52, 311–317.